

Colossal Magnetoresistance in Focus: Studies of Different CMR Materials by Mössbauer Spectroscopy

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The study of transition metal based materials displaying colossal magnetoresistance (CMR) is one of today's most exciting fields of research mainly due to the scientific challenge of finding the possibly common root (if that exists) of the CMR effect in the various materials. In this review we present novel results obtained by ⁵⁷Fe Mössbauer studies done on three different types of CMR materials, namely the cobalt based perovskites $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$, the $\text{Sr}_2\text{FeMoO}_6$ double perovskite, and copper containing chalcogenide spinels ($\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$).

1. Introduction

Magnetoresistance (MR) is the change of the electrical resistivity due to an applied external magnetic field. It is usually expressed as $MR = [R(H) - R(0)]/R(0)$ where $R(0)$ and $R(H)$ denote the electrical resistivity measured without and with H external magnetic field applied, respectively. In the case of negative magnetoresistance the value of MR is restricted to the range of -100% to 0% . Although magnetoresistance was known long ago, the technological applicability of the MR effect was only available when giant negative magnetoresistance (GMR) was found in magnetic multilayers even at room temperature. These materials are used nowadays in the read-heads of several magnetic storage devices. The discovery of an even higher MR in manganese perovskites (called colossal magnetoresistance, CMR)¹ raised renewed interest in the transition metal based materials with anomalous MR properties.

In the present article we report the results of some recent investigations on members of three different classes of CMR materials: $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ perovskites, $\text{Sr}_2\text{FeMoO}_6$ double perovskite, and $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ chalcogenide spinels.

The magnetoresistance properties of the investigated materials are demonstrated in Figure 1. The figure illustrates well the variety of magnetoresistance phenomena in materials with different structure. While the $\text{Fe}_x\text{Cu}_{1-x}\text{Cr}_2\text{S}_4$ spinels show a CMR effect² similar to that observed originally in the manganese perovskites, in the double perovskite system $\text{Sr}_2\text{FeMoO}_6$, MR decreases near linearly with temperature,³ and the iron doped cobalt perovskites display huge negative MR below ~ 50 K as well as above ~ 150 K.⁴

Iron is thought to play a key role in the realization of magnetoresistance in each of the investigated compounds. Consequently, the study of the local state of iron as a function of temperature, sample composition, etc. may reveal the microscopic background of the magnetoresistance effect in these systems. As ⁵⁷Fe Mössbauer spectroscopy can reflect even

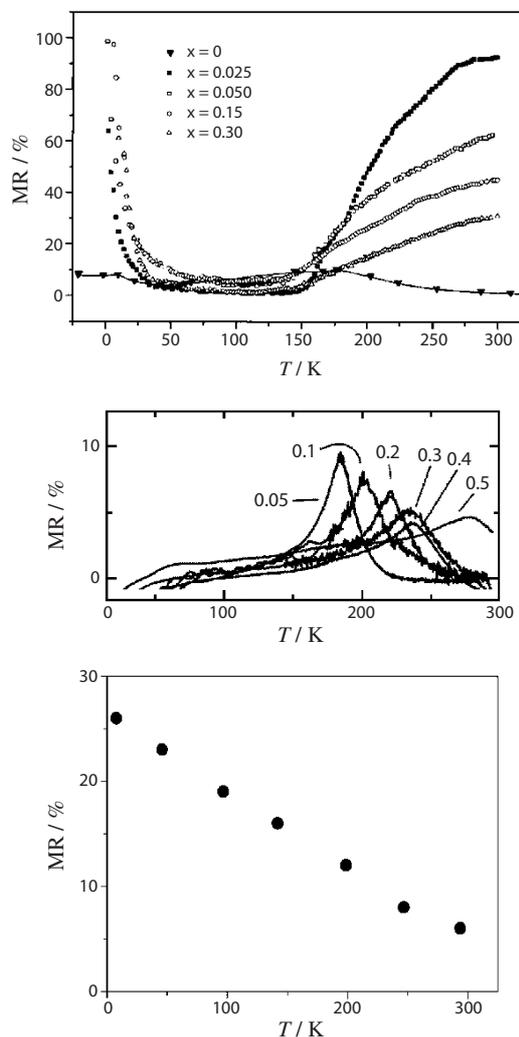


Figure 1. Magnetoresistance properties of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ perovskites⁴ (top, the magnetic field was 7.5 T for $x > 0$ and 7 T for $x = 0$), $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ chalcogenide spinels² (in the middle, the numbers inside the picture correspond to the copper concentration, x , the applied magnetic field was 5 T), and $\text{Sr}_2\text{FeMoO}_6$ double perovskite³ (bottom, measured in a magnetic field of 0.4 T).

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minute changes in the local electronic, magnetic, and vibrational state of iron, it is an especially useful tool in the study of these materials.

2. $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ Perovskites

While in manganese perovskites the colossal magnetoresistance effect is observed only at temperatures in the neighborhood of the Curie temperature, iron doped $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ ($0.025 < x < 0.3$) display negative magnetoresistance that increases monotonically with temperature in the temperature interval $150 \text{ K} \leq T \leq 300 \text{ K}$. Additionally, $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ also displays strong negative MR below $T \approx 50 \text{ K}$.⁴ The negative MR observed above $T \approx 150 \text{ K}$ is particularly interesting because, unlike in the case of manganites, in $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ the magnitude of MR increases continuously even in the paramagnetic phase. Namely, the Curie-temperature of the iron free end member $\text{La}_{0.8}\text{Sr}_{0.2}\text{CoO}_{3-d}$ is $T_C \approx 180 \text{ K}$,⁵ and T_C decreases with the x iron concentration.⁶

Above $T \approx 150 \text{ K}$, the amplitude of negative magnetoresistance in $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ perovskites was found to decrease with increasing iron concentration in the range of 0.025 to 0.3. At the same time, the iron free $\text{La}_{0.8}\text{Sr}_{0.2}\text{CoO}_{3-d}$ perovskite displays only a modest magnetoresistance effect around its Curie-temperature. This suggests that iron plays a key role in the realization of the considerable negative magnetoresistance observed in $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_x\text{Co}_{1-x}\text{O}_{3-d}$ perovskites.^{4,7}

One of the main questions regarding these perovskites is how already a low amount of iron substituted for Co manages to change dramatically the MR properties of the mother compound $\text{La}_{0.8}\text{Sr}_{0.2}\text{CoO}_{3-d}$. To collect information in this regard, we decided to take a closer look on the temperature dependence of the local state of iron in $\text{La}_{0.8}\text{Sr}_{0.2}^{57}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$. The local electronic, magnetic and vibrational state of iron can be advantageously studied by the help of ^{57}Fe transmission and ^{57}Co (^{57}Fe) emission Mössbauer spectroscopy (in the followings referred to as TMS and EMS, respectively).

The $\text{La}_{0.8}\text{Sr}_{0.2}^{57}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ sample was prepared by the following method. At first, $\text{La}_{0.8}\text{Sr}_{0.2}\text{CoO}_3$ samples were fabricated via the citric acid-ethylene glycol method. The stoichiometric amounts of $\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$, SrCO_3 and $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ were dissolved in 50 mL of concentrated nitric acid. ^{57}Fe metal was dissolved in nitric acid and the respective aliquot was added to the nitric acid solution. The solution was diluted with 200 mL water. Thirty grams of citric acid and 8 g of ethylene glycol were added. The aqueous solution was evaporated until a brown gel was obtained. The gels were dried under vacuum at 125°C . The residues were ground, first heated to 500°C , kept this temperature for two hours and then heated to 800°C . The heating rates were 5 K min^{-1} . The samples were kept at 800°C for five hours. Upon ballistic cooling the samples were ground and compacted at 750 MPa into discs of 10 mm diameter. The specimens were then sintered in air at 1200°C for 24 hours. Heating rates were 3 K min^{-1} , cooling rates 10 K min^{-1} .

^{57}Fe transmission Mössbauer spectroscopy measurements were carried out on powdered samples. The temperature dependent ^{57}Fe TMS measurements were performed in a temperature controlled flow-through type liquid nitrogen cryostat (Leybold) and an APD type liquid He cooled cryostat. During the measurements the temperature of the samples was kept constant with a precision of $\Delta T \approx \pm 0.5 \text{ K}$. The γ -rays were provided by a $^{57}\text{Co}(\text{Rh})$ source with 10^9 Bq activity.

For the purpose of ^{57}Co (^{57}Fe) emission Mössbauer measurements, the $\text{La}_{0.8}\text{Sr}_{0.2}^{57}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ sample was doped by ^{57}Co . Ethanolic solution of 100 MBq carrier-free ^{57}Co -nitrate was transferred onto the La-Sr-cobaltate pellet drop-by-drop, and after the evaporation of the alcohol, the pellet was treated for 3 hours in dry oxygen flow at 1000°C for a perfect isotopic exchange. XRD measurement confirmed that the perovskite

structure and the composition of the original sample were not changed during the associated heat treatment. Two Mössbauer spectra using the conventional emission geometry⁸ were recorded at 4.2 K (in a helium bath cryostat) and at room temperature.

^{57}Fe isomer shift values are given relative to α -iron at room temperature. Mössbauer spectra were analyzed with version 3.0 of the MossWinn program.⁹

^{57}Fe transmission Mössbauer spectra of $\text{La}_{0.8}\text{Sr}_{0.2}^{57}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ at different temperatures are shown in Figure 2. Above $T \approx 45 \text{ K}$ the spectrum consists of a quadrupole doublet with a small quadrupole splitting (Δ) of $\Delta \approx 0.2 \text{ mm s}^{-1}$. The isomer shift (δ) of the doublet increases from $\delta \approx 0.33 \text{ mm s}^{-1}$ at $T = 290 \text{ K}$ to $\delta \approx 0.4 \text{ mm s}^{-1}$ at $T = 80 \text{ K}$. These δ and Δ values indicate that the electronic state of iron is high spin Fe^{3+} ($t_{2g}^3e_g^2$) in this compound, and in contrast to some assumptions (see, e.g., Reference 10) it does not support the existence of Fe^{4+} in this system. The Lorentzian curves of the quadrupole doublet are rather broad; their line width (Γ) is around $\Gamma \approx 0.35 \text{ mm s}^{-1}$ at $T = 290 \text{ K}$ and it increases with decreasing temperature.

In the spectrum taken at $T = 45 \text{ K}$ the quadrupole doublet is superimposed on a strongly broadened absorption curve (Figure 2). This latter broadened subspectrum originates from iron cations whose magnetic moment displays a slowed-down relaxation rate compared to the iron cations contributing to the paramagnetic spectrum component still observable in the Mössbauer spectrum. Namely, when the relaxation time of the magnetic moment of iron becomes comparable to the Larmor precession time ($\tau_L \approx 10^{-7} \text{ s}$) of the ^{57}Fe nucleus, then the magnetic hyperfine interaction, although still being averaged out to some extent, becomes observable in the ^{57}Fe Mössbauer spectrum. A further slow-down of magnetic relaxation with decreasing temperature results in the appearance of a full-developed sextet as clearly observed in the spectrum taken at $T = 10 \text{ K}$.

The detailed temperature dependence of the δ , Δ , and Γ ^{57}Fe Mössbauer parameters of $\text{La}_{0.8}\text{Sr}_{0.2}^{57}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ are shown

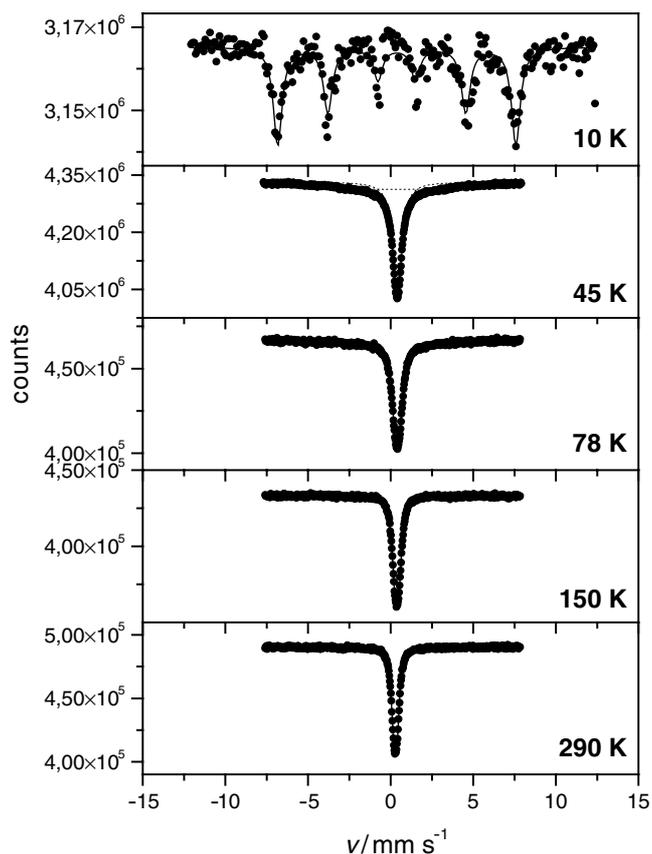


Figure 2. ^{57}Fe transmission Mössbauer spectra of $\text{La}_{0.8}\text{Sr}_{0.2}^{57}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ at selected temperatures.

in Figure 3.

The ^{57}Fe isomer shift decreases linearly with temperature from 78 K to ~ 150 K and from ~ 160 K to room temperature, but at ~ 150 K the slope of the decline changes.

The gradient $\frac{d\delta(T)}{dT}$ is inversely proportional to the effective vibrating mass (M_{vibr}) of the iron cations.¹¹ The magnitude of $\frac{d\delta(T)}{dT}$ increases suddenly at ~ 150 K, which is equivalent to a sudden decrease in M_{vibr} with increasing temperature. Such a decrease is to be expected when the vibrational motion of iron and that of the neighboring cations becomes less correlated due to a decrease in the force that made them move in a more correlated manner previously. Thus a sudden decrease in the strength of the chemical bonding or in the strength of the magnetic interaction between iron and its surroundings may well result in a sudden decrease of M_{vibr} . The ratio $M_{\text{vibr}}(T > 150 \text{ K})/M_{\text{vibr}}(T < 150 \text{ K}) \approx 2/3$ indicates that the vibrational state of iron changes considerably at $T \approx 150$ K in $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$.

The Mössbauer line width of the quadrupole doublet starts to rise (with decreasing temperature) at around the same temperature ($T \approx 150$ K) where the effective vibrating mass changes suddenly (Figure 3). The evaluated ^{57}Fe quadrupole splitting (Figure 3) also increases with decreasing temperature in correlation with the line width.

The temperature $T \approx 150$ K at the same time coincides with the ferromagnetic ordering temperature (T_C) of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$.⁶ Thus it is plausible to assume that the sudden decrease of M_{vibr} at $T \approx 150$ K is a consequence of the FM \rightarrow PM transition of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$. Similarly, the rather anomalous increase of Γ and Δ Mössbauer parameters with decreasing temperature below $T \approx 150$ K is likely to be of magnetic origin. The increase of the Mössbauer line width reflects that the relaxation time of the iron magnetic moment increases to around 10^{-8} s when the Co host matrix orders magnetically. Such an increased relaxation time is expected to occur when

iron magnetic moments are relaxing not individually like in a paramagnet, but they are part of nanosized magnetic clusters as is the case, e.g., in superparamagnetic materials. When the relaxation time increases over the Larmor precession time of the ^{57}Fe nucleus, then a full-developed magnetic spectrum is observed as displayed in Figure 2. At this point one should note that the apparent increase in the Δ parameter with decreasing temperature (Figure 3) below $T \approx 150$ K is not necessarily connected to a change in the quadrupole interaction, but it may also be the result of the slowing down of magnetic relaxation that appears in the spectrum as an apparent increase in the separation of the two broadened lines that we fit as a doublet.

Figure 4 shows the $^{57}\text{Co}(^{57}\text{Fe})$ EMS spectrum of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ at temperatures 4.2 K and 290 K. The results obtained in this way are basically the same as obtained by the TMS technique (Figure 2). This shows that in this compound the electronic relaxation time of the nucleogenic iron is much shorter than the lifetime of the excited Mössbauer nuclear level of the ^{57}Fe nucleus ($\sim 10^{-7}$ s), and as a consequence the Mössbauer parameters observed via the EMS measurement also reflect the equilibrium chemical state of iron cations. In this case, however, as the nucleogenic iron is formed at the site of a Co cation, the values we obtain for the ^{57}Fe Mössbauer parameters represent the state of iron cations situated at *Co sites* in $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$. The good agreement between the ^{57}Fe Mössbauer parameters obtained via TMS and EMS techniques shows that in $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ the substituted 5 mol% ^{57}Fe does not form iron clusters, but it is distributed (probably near uniformly) in the Co host matrix. Thus, considering also the relatively low amount of ^{57}Fe present in the system, the magnetic state of iron has to be determined solely by Fe-Co magnetic exchange interactions.

The observation of magnetic relaxation in the ^{57}Fe Mössbauer spectra below the magnetic ordering temperature indicates that the magnetic moment of iron is strongly connected to and relaxes together with magnetically ordered nanosized Co clusters. This indicates that the magnetic exchange interaction between Fe and Co is strong in this system, and consequently when the material gets ordered magnetically at $T \approx 150$ K iron magnetic moments get also ordered (or are already ordered) with respect to neighboring Co magnetic moments. As a consequence, the relaxation behavior we observe in the ^{57}Fe Mössbauer spectra of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ must correspond to the magnetic relaxation of nanosized Co clusters existing in this perovskite. This result further confirms the idea¹² that the existence of nanoscale magnetic clusters is a prerequisite of intrinsic negative magnetoresistivity.

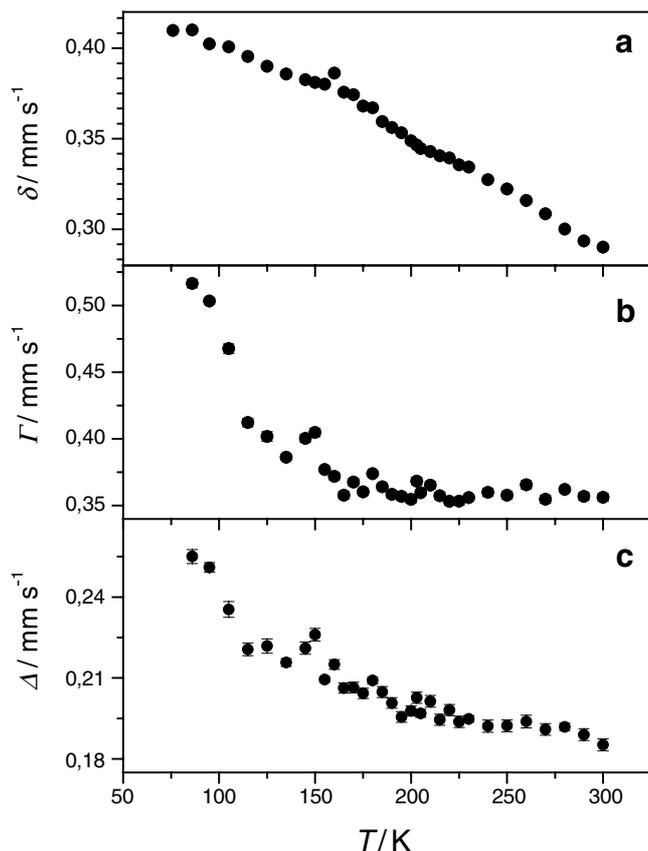


Figure 3. The temperature dependence of ^{57}Fe Mössbauer isomer shift (δ), line width (Γ), and quadrupole splitting (Δ) of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$.

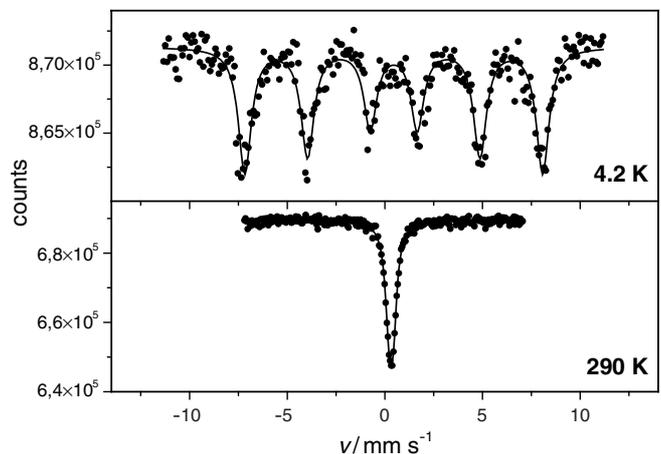


Figure 4. $^{57}\text{Co}(^{57}\text{Fe})$ emission Mössbauer spectra of $\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$ at temperatures 4.2 K and 290 K.

3. Sr₂FeMoO₆: A Double Perovskite System

The double perovskites Sr₂FeMoO₆ has been brought recently into the center of scientific interest because of their considerable magnetoresistance observed already in relatively low magnetic fields even at and above room temperature.^{13–19}

The double perovskite Sr₂FeMoO₆ has been reported to have the cubic ABO₃ structure that contains Fe and Mo cations ordered alternately at the B site.¹⁴

Sr₂FeMoO₆ has long been known as a conducting ferrimagnet with a relatively high magnetic transition temperature of $T_c \approx 450$ K.²⁰ Recently, however, neutron diffraction measurements revealed zero magnetic moment on Mo cations, and consequently the ferrimagnetic structure of Sr₂FeMoO₆ was disputed.¹⁴ It was suggested, that while localized magnetic moment can be found only on ferromagnetically arranged Fe³⁺ cations, the 4d¹ electron of Mo is delocalized as well as spin polarized due to the involvement of Fe³⁺ t_{2g} levels in the conduction band, the latter being formed by the hybridization of 3d (Fe) and 4d (Mo) electronic states.^{14, 16–18}

At the same time, in Sr₂FeMoO₆ saturation magnetic moments of 3.0 μ_B,¹³ 3.6 μ_B,³ 3.18 μ_B,²⁰ and 3.08 μ_B²¹ were observed at low temperatures on the basis of magnetization measurements. These values are below the range from 4 μ_B to 5 μ_B expected for Fe^{+(3-δ)}, where δ ≤ 1 denotes the contribution of the delocalized electron to the Fe minority spin t_{2g} level. The decreased value of saturation magnetic moment has been attributed to a deviation from the perfectly ordered structure by the replacement of a certain amount of Mo by Fe and vice versa. Indeed, it was shown that iron substituting Mo aligns antiferromagnetically to neighboring iron cations in Sr₂FeMoO₆.²² Recently it was suggested that this kind of anti-site defects might be responsible for low field magnetoresistance observed in polycrystalline Sr₂FeMoO₆.^{23, 24}

In order to gain insight into the effects of cation disorder on the local electronic structure, in this article we studied a double perovskite Sr₂FeMoO₆ that contains a considerable amount of misplaced cations.

The studied Sr₂FeMoO₆ sample was prepared in the following way. Stoichiometric mixture of SrCO₃, Fe₂O₃, and MoO₃ was homogenized and pressed into pellet under a pressure of 100 MPa. The pellet was calcined in air at 900°C for 4 hours. The precursor obtained in this way, consisting of SrMoO₄ and SrFeO₃, was grinded and pressed again into a pellet. This procedure was followed by a heat treatment at 900°C for 2 hours under nitrogen atmosphere containing 5

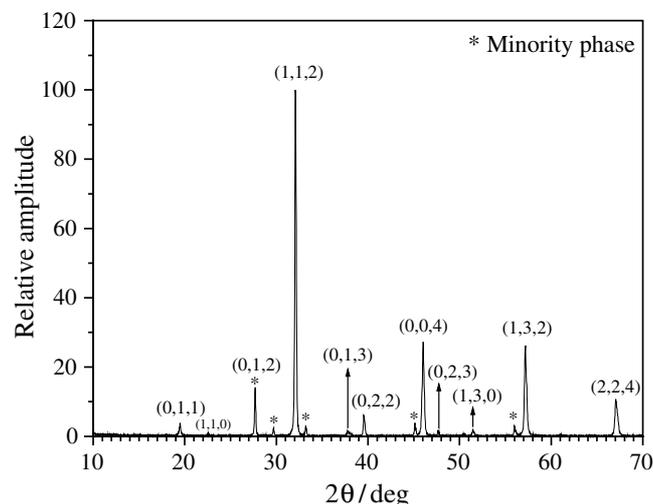


Figure 5. X-ray diffractogram of Sr₂FeMoO₆ powder measured with Cu K_α ($\lambda = 0.154184$ nm) radiation at room temperature. The (*hkl*) indexing of the peaks is shown for the *I4/m* tetragonal structure. The peaks belonging to the minority phase are marked by * sign. The strongest line of the minority phase coincides with the (012) line of the main phase.

(V/V)% H₂. The heat treatment was repeated once after regrinding and repressing the sample.

Figure 5 shows the X-ray diffractogram of the investigated Sr₂FeMoO₆ sample. The diffractogram can be decomposed into two components. The crystal structure of the main component can be identified as a double perovskite with a tetragonal *I4/m* space group and a lattice parameter of $a = 0.5586$ nm and $c = 0.7888$ nm. This result is in agreement with the earlier neutron diffraction investigations¹⁹ indicating that below 420°C the *I4/m* tetragonal and above this temperature the cubic *Fm3m* structure is stable.

The Bragg peaks not belonging to the double perovskite structure originate from a simple tetragonal minority phase with *I4/a* space group and with cell parameters of $a = 0.53943$ nm and $c = 1.20436$ nm. The volume fraction of this simple tetragonal phase, calculated on the basis of the ratio of integrated Bragg line intensities, was estimated to be around 6%. The cell parameters of this minority phase are very close to that of the precursor material SrMoO₄ (*I4/a* tetragonal, $a = 0.53944$ nm, $c = 1.2020$ nm).

Figure 6 shows the ⁵⁷Fe Mössbauer spectrum of Sr₂FeMoO₆ measured at $T = 4.2$ K. The spectrum can be decomposed into four different magnetically split subspectra reflecting magnetically ordered iron cations with different electronic configurations on their 3d level. In the followings, these subspectra will be referred to as S1 (61%), S2 (13%), S3 (15%), and S4 (11%). The ⁵⁷Fe isomer shift and hyperfine magnetic field (*B*) parameters belonging to the observed subspectra are visualized in Figure 7 together with the corresponding electronic configu-

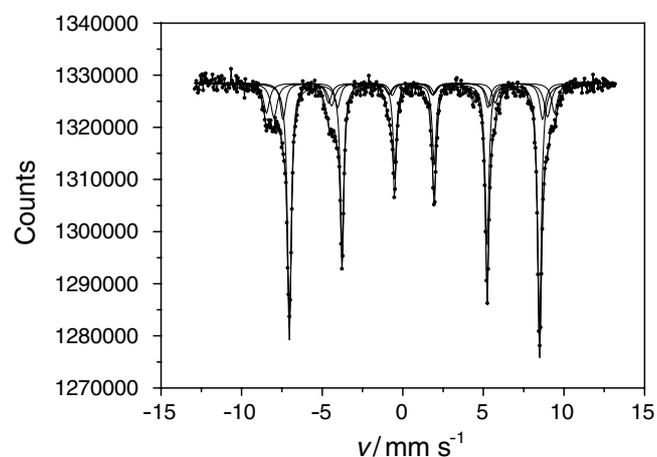


Figure 6. ⁵⁷Fe Mössbauer spectrum of Sr₂FeMoO₆ at $T = 4.2$ K, and its decomposition into four magnetically split subspectra.

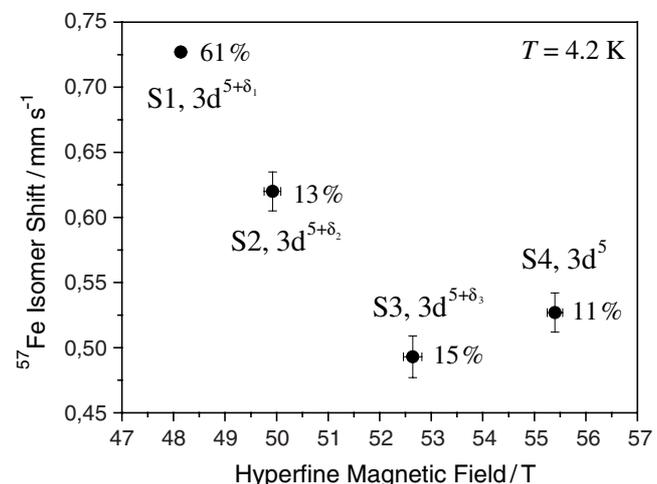


Figure 7. ⁵⁷Fe isomer shift and hyperfine magnetic field Mössbauer parameters obtained for the four magnetic subcomponents observed in the ⁵⁷Fe Mössbauer spectrum of Sr₂FeMoO₆ at $T = 4.2$ K. The suggested electronic configurations of the corresponding iron cations are indicated next to the data points. It is assumed that $\delta_3 < \delta_2 < \delta_1$.

ration of the iron cations. The main sextet S1 can be attributed to the ideal, perfectly ordered regions of the double perovskite. Its Mössbauer parameters agree well with those reported in Reference 19 for $\text{Sr}_2\text{FeMoO}_6$. While the corresponding isomer shift value is relatively high, the hyperfine magnetic field is relatively low compared to the values expected for high spin Fe^{3+} at $T = 4.2$ K ($\delta \approx 0.5$ mm s $^{-1}$, $B \approx 55$ T). This points to a considerable electron density on the minority spin t_{2g} level of iron, which is in agreement with a high degree of hybridization between 3d(Fe) and 4d(Mo) electronic levels. Furthermore, the intermediate value of isomer shift and hyperfine magnetic field Mössbauer parameters, i.e. between values that would be representative for Fe^{3+} and Fe^{2+} , indicates that the δ_1 minority spin electron density at the $3d^{5+\delta_1}$ level of iron (Figure 7) arises from delocalized electrons. The S2 magnetic component in the ^{57}Fe Mössbauer spectrum was also detected earlier in $\text{Sr}_2\text{FeMoO}_6$.¹⁹ This component displays a decreased isomer shift and an increased hyperfine magnetic field compared to the case of the perfectly ordered structure represented by S1 (Figure 7). This corresponds to a decreased electron density (δ_2) at the minority spin $3d^{5+\delta_2}$ level of iron. The decreased minority spin electron density can be interpreted as a reduced degree of delocalization of the $4d^1$ electron of Mo.

In contrast with earlier ^{57}Fe Mössbauer studies done on $\text{Sr}_2\text{FeMoO}_6$ with moderate levels of cation disorder, in the Mössbauer spectrum of our $\text{Sr}_2\text{FeMoO}_6$ we observe two more subspectra, S3 and S4, whose hyperfine magnetic field is higher and whose ^{57}Fe isomer shift is lower than the corresponding values representing S1 and S2. Thus, S3 and S4 can be attributed to an electronic state of iron where the degree of delocalization of the minority spin electron is further reduced compared to the case of S1 and S2. Actually, the ^{57}Fe Mössbauer parameters of the component S4 (Figure 7) already refer to a fully localized state of high spin Fe^{3+} with the $3d^5$ electronic configuration. Although the isomer shift of the species S3 equals that of S4 within the experimental uncertainty, its decreased hyperfine magnetic field already refers to the existence of moderate minority spin electron density on the $3d^{5+\delta_3}$ level of the corresponding iron cations (Figure 7). The less pronounced difference between the isomer shift of S3 and S4 can be explained by assuming that beside the 3d(Fe) and 4d(Mo) electronic states the 4s(Fe) electronic state of iron is also involved in some extent in the hybridization of electronic levels forming the conduction band.

The relatively high extent of cation disorder expected to be present in our sample is thought to be responsible for the appearance of the two additional sextets, S3 and S4. This means that cation disorder results in a decreased delocalization of the $4d^1$ electron of Mo. Mislaced cations are expected to form barriers to delocalized electrons. For example, an antisite iron is expected to repel itinerant electrons of spin polarized nature because of its antiferromagnetic alignment relative to neighboring iron cations. Thus, cation disorder leads to electrically insulating domains in the otherwise half metallic double perovskite. The degree of electronic insulation at the same time is expected to depend on the size of the insulating domains. While the electrons may enter into the domain boundaries and may also pass through thin insulating domains, they are completely excluded from the inside of bulkier domains. Accordingly, in the presence of domains with considerable volume fraction of cation disorder, iron cations belonging to different levels of minority spin electron density are expected to exist in the double perovskite. This explains the appearance of the magnetic subcomponents S2, S3, and S4 in the ^{57}Fe Mössbauer spectrum of $\text{Sr}_2\text{FeMoO}_6$. Namely, while S2 can be attributed to moderate disorder levels (e.g. isolated antisite defects), S3 and S4 may arise from iron cations located in the inside of bulkier insulating domains.

4. $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ Type Spinel

Apart from perovskites, recently chalcogenide spinels were also found showing remarkable CMR properties.^{2, 25–27} Although Mössbauer studies of some iron containing chalcogenides were already performed several decades ago (see, e.g., Reference 28), we present some new ^{57}Fe Mössbauer measurements in order to gain a deeper insight into the structure of $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ type chalcogenides.

The $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ ($x = 0, 0.25, 0.5$, and 0.75) specimens were prepared by solid state reactions. The stoichiometrically mixed samples sealed in an evacuated quartz tube were fired for 12 h at 850°C , and then they were ground and pressed into pellets. Sample purity was checked by X-ray diffraction and by scanning electron microscopy in combination with energy dispersive X-ray fluorescence spectroscopy.²⁹ ^{57}Fe Mössbauer spectroscopy measurements were carried out on the $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ samples in transmission geometry with a conventional constant acceleration type spectrometer. ^{57}Fe isomer shifts are given relative to α -iron at room temperature.

The temperature dependence of the ^{57}Fe Mössbauer parameters characteristic for iron at the tetrahedral site in the sample FeCr_2S_4 ($x = 0$) was published earlier.^{30,31} The Curie temperature (T_c) was found to be about 170 K and the magnetic field was 20 T in this sample at 76 K. The isomer shift (δ) was 0.72 mm s $^{-1}$ at 76 K and it decreased to 0.6 mm s $^{-1}$ when the temperature was increased to room temperature. These parameters confirm the oxidation level of iron(II).³⁰

Figure 8a and b show the ^{57}Fe Mössbauer spectra of the samples $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ (for $x = 0, 0.25, 0.5$, and 0.75) recorded at room temperature. As clearly reflected by the ^{57}Fe

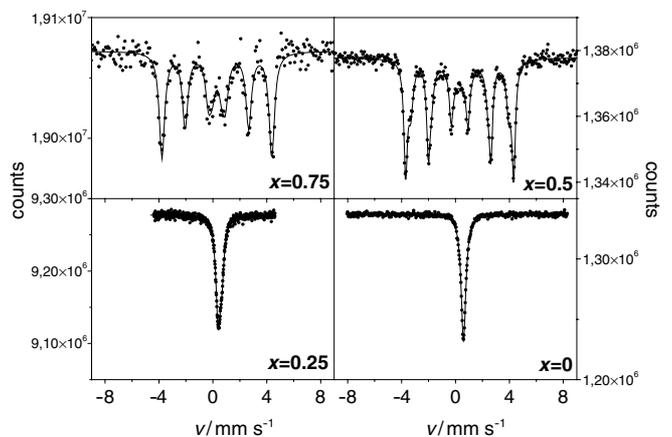


Figure 8a. ^{57}Fe Mössbauer spectra of $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ ($x = 0, 0.25, 0.5$, and 0.75) at room temperature (only the envelope of the fitting curves are shown).

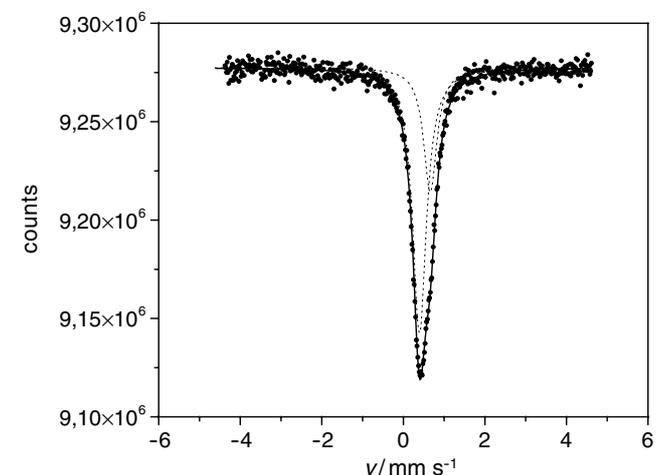


Figure 8b. ^{57}Fe Mössbauer spectrum of $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ at room temperature showing the decomposition of the absorption curve into two singlets belonging to paramagnetic Fe^{3+} and Fe^{2+} .

Mössbauer spectrum shown in Figure 8b, already a substitution of 25% of iron with Cu results in considerable differences in the oxidation state of iron in these compounds. While in FeCr_2S_4 all iron cations display the same (Fe^{2+}) oxidation state, the ^{57}Fe Mössbauer spectrum of $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ consists of two clearly distinguishable subspectra each revealing a different oxidation state of iron in the chalcogenide. The singlet with Mössbauer isomer shift $\delta = 0.66(1) \text{ mm s}^{-1}$ can be assigned to Fe^{2+} state that has been reported in FeCr_2S_4 .³⁰ The second singlet with isomer shift $\delta = 0.40(1) \text{ mm s}^{-1}$, however, undoubtedly refers to the existence of Fe^{3+} cations in the $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ sample. Considering the requirement of charge neutrality, and assuming that the concentration of sulfur vacancies is negligible compared to the concentration of Cu, it follows that at least part of the Cu being present in the system must be in the Cu^+ state, then one would expect the relative amount of Fe^{2+} and Fe^{3+} to be $\text{Fe}^{2+} : \text{Fe}^{3+} = 2 : 1$. The room temperature ^{57}Fe Mössbauer spectrum of $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ is, however, not in agreement with this expectation (Figure 8b). Namely, by comparing the peak areas belonging to Fe^{2+} and Fe^{3+} one observes a ratio of $\text{Fe}^{2+} : \text{Fe}^{3+} \approx 1 : 2.2$. One should consider, however, that the ratio of Mössbauer peak areas reflects faithfully the relative occurrence of the corresponding iron species only if the recoilless fraction (the probability of the Mössbauer effect, also called the Mössbauer-Lamb factor) is the same for the two species. As the recoilless fraction should have a tendency to approach 100% (from below) with decreasing temperature for all species, a Mössbauer measurement carried out at a lower temperature should result in a Mössbauer spectrum in which the area ratio of the peaks reflects the relative occurrence of the corresponding iron species more faithfully than it does in the case of the measurement done at room temperature (Figure 8a). For this reason we also carried out a ^{57}Fe Mössbauer measurement of $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ at $T = 77 \text{ K}$, which resulted in the Mössbauer spectrum shown in Figure 9. The Mössbauer spectrum at $T = 77 \text{ K}$ can be decomposed into three different components. The component belonging to Fe^{2+} is now a magnetically split sextet ($\delta = 0.67(2) \text{ mm s}^{-1}$, $B =$

$22.9(2) \text{ T}$) showing that at $T = 77 \text{ K}$ Fe^{2+} cations are already ordered magnetically in agreement with results obtained earlier for FeCr_2S_4 .^{30, 31} At the same time we observe two different subspectra with parameters representative for Fe^{3+} . One of them is a magnetically split sextet ($\delta = 0.52(1) \text{ mm s}^{-1}$, $B = 33.1(1) \text{ T}$) showing that there are also Fe^{3+} cations magnetically ordered at $T = 77 \text{ K}$, whereas the other is a quadrupole doublet ($\delta = 0.55(2) \text{ mm s}^{-1}$, $\Delta = 1.19(3) \text{ mm s}^{-1}$) referring to Fe^{3+} cations in the paramagnetic state. The relative area fraction of the latter component is $\sim 9\%$. The nearly equal isomer shift of the two different Fe^{3+} components suggests that the corresponding cations may be situated in the same crystallographic position in $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$, but the iron cations belonging to the doublet are probably separated from the magnetically ordered regions (e.g. by sulfur vacancies) and consequently they display paramagnetic behavior at $T = 77 \text{ K}$. At the same time it is also not entirely excluded that the paramagnetic Fe^{3+} subspectrum at $T = 77 \text{ K}$ originates from some FeS_x impurities in the sample.

By analyzing the area ratio of subspectra belonging to Fe^{2+} and Fe^{3+} cations in the Mössbauer spectrum taken at $T = 77 \text{ K}$ one observes $\text{Fe}^{2+} : \text{Fe}^{3+} \approx 3 : 2$ which ratio is closer to the expected ratio $2 : 1$ than that observed at room temperature ($1 : 2.2$). The significant difference between the apparent $\text{Fe}^{2+} : \text{Fe}^{3+}$ ratios observed at $T = 290 \text{ K}$ and at $T = 77 \text{ K}$ indicates that in $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ at room temperature the Mössbauer-Lamb factor belonging to Fe^{3+} is at least three times greater than that belonging to Fe^{2+} . This result shows that by replacing iron with copper in $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ it is not only the oxidation state of iron that changes, but simultaneously with the transition $\text{Fe}^{2+} \rightarrow \text{Fe}^{3+}$ the local vibrational state of iron cations is altered considerably, too.

In contrast with $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$, the compounds $\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$ and $\text{Fe}_{0.25}\text{Cu}_{0.75}\text{Cr}_2\text{S}_4$ are magnetically ordered already at room temperature as revealed by the corresponding ^{57}Fe Mössbauer spectra in Figure 8.

The Mössbauer spectrum of $\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$ can be decomposed into a magnetically split sextet (with parameters $\delta = 0.31(1) \text{ mm s}^{-1}$, $B = 24.5(1) \text{ T}$) and a quadrupole doublet (with

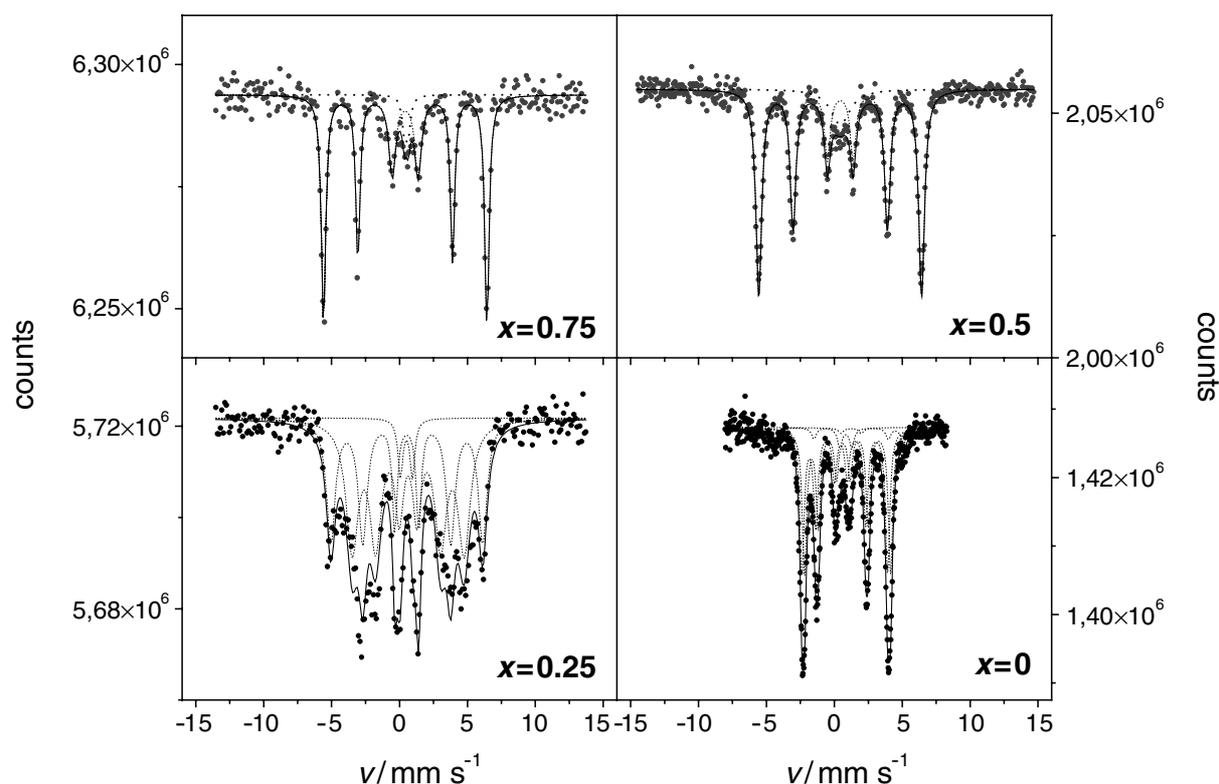


Figure 9. ^{57}Fe Mössbauer spectra of $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ ($x = 0, 0.25, 0.5,$ and 0.75) at $T \approx 77 \text{ K}$.

TABLE 1: ^{57}Fe Mössbauer Parameters of $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ ($x = 0, 0.25, 0.5, \text{ and } 0.75$) Chalcogenide Spinel

Sample T/K	FeCr_2S_4		$\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$		$\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$		$\text{Fe}_{0.25}\text{Cu}_{0.75}\text{Cr}_2\text{S}_4$	
	75 K	290 K	78 K	290 K	76 K	290 K	77 K	290 K
$\text{Fe}^{2+} / \%$	100	100	59	31	–	–	–	–
$\delta / \text{mm s}^{-1}$	0.71	0.59	0.65	0.66	–	–	–	–
B / T	20.0	–	22.6	–	–	–	–	–
$\text{Fe}^{3+} (\text{M}) / \%$	–	–	32	–	88	90	94	82
$\delta / \text{mm s}^{-1}$	–	–	0.51	–	0.43	0.31	0.42	0.30
B / T	–	–	33.0	–	37.2	24.5	37.3	25.3
$\text{Fe}^{3+} (\text{PM}) / \%$	–	–	9	69	12	10	6	18
$\delta / \text{mm s}^{-1}$	–	–	0.55	0.40	0.45	0.3	0.50	0.3
$\Delta / \text{mm s}^{-1}$	–	–	(1.2)	–	(0.5)	(1.0)	(0.29)	(0.9)

' $\text{Fe}^{3+} (\text{M})$ ' and ' $\text{Fe}^{3+} (\text{PM})$ ' denote magnetically ordered Fe^{3+} and Fe^{3+} in a paramagnetic state, respectively. Quadrupole splitting (Δ) values given in parentheses are uncertain because of the strong overlap of the absorption lines. Isomer shift (δ) and magnetic field (B) Mössbauer parameters concerning Fe^{2+} in FeCr_2S_4 at $T = 75$ K are average values, see Reference 30 for details.

parameters $\delta = 0.31(3) \text{ mm s}^{-1}$, $\Delta = 0.9(1) \text{ mm s}^{-1}$).

The relative area fraction of the doublet is $\sim 10\%$. While at room temperature the two peaks of the doublet overlap with the inner two lines of the sextet, at $T \approx 76$ K the existence of the doublet becomes more obvious (Figure 9). The origin of this doublet is likely to be the same as that found in the case of $\text{Fe}_{0.75}\text{Cu}_{0.25}\text{Cr}_2\text{S}_4$ at $T = 77$ K, namely Fe^{3+} in the neighborhood of a sulfur vacancy. This assignment is supported by the ^{57}Fe Mössbauer spectrum of $\text{Fe}_{0.25}\text{Cu}_{0.75}\text{Cr}_2\text{S}_4$ at $T = 290$ K (Figure 8) which also contains a doublet with the same isomer shift and quadrupole splitting parameters, but the relative area fraction of this doublet is 18% which is a considerably increased value compared to the case of $\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$. This increase can be understood again considering the requirement of macroscopic charge neutrality. Namely, the isomer shift of the main magnetic component in the ^{57}Fe Mössbauer spectra of $\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$ and $\text{Fe}_{0.25}\text{Cu}_{0.75}\text{Cr}_2\text{S}_4$ undoubtedly refers to Fe^{3+} , which further supports that copper is present mainly in the Cu^+ state in this system. It follows that without any sulfur vacancies in $\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$ the capability of the iron subsystem to compensate for the presence of Cu^+ cations by the $\text{Fe}^{2+} \rightarrow \text{Fe}^{3+}$ transition (in order to maintain charge neutrality) is already fully exhausted. Therefore, if we replace more iron with Cu^+ in the system, the missing positive charge has to be compensated by the appearance of sulfur vacancies (i.e. missing negative charges). And, assuming that the relative area fraction of the observed doublet component is really proportional to the number of sulfur vacancies, this is exactly what we observe in the ^{57}Fe Mössbauer spectra. (It should be noted here, that our reasoning remains valid even if a certain amount of copper is in the Cu^{2+} state in $\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$ with $x \geq 0.5$.)

Although the ^{57}Fe isomer shift $\delta \approx 0.3 \text{ mm s}^{-1}$ found for the chalcogenides with $x \geq 0.5$ clearly refers to Fe^{3+} , the hyperfine magnetic field reflected by the main magnetic component in the ^{57}Fe Mössbauer spectrum of $\text{Fe}_{0.5}\text{Cu}_{0.5}\text{Cr}_2\text{S}_4$ and $\text{Fe}_{0.25}\text{Cu}_{0.75}\text{Cr}_2\text{S}_4$ (Table 1) is much lower than the typical magnetic fields in the high spin iron(III) compounds (45–55 T at 76 K). This contradiction between the measured isomer shift and magnetic field values can be resolved if we take into account that there is a covalent bond between sulphur and iron and suppose that the electronic structure of iron in this sample is: $3d^{5+x}4s^y$ with $x > 0$, $y > 0$.

5. Conclusions

Iron is thought to play a decisive role in determining the magnetic state and the MR properties of the studied perovskite ($\text{La}_{0.8}\text{Sr}_{0.2}\text{Fe}_{0.05}\text{Co}_{0.95}\text{O}_{3-d}$), double perovskite ($\text{Sr}_2\text{FeMoO}_6$),

and chalcogenide spinel ($\text{Fe}_{1-x}\text{Cu}_x\text{Cr}_2\text{S}_4$) systems. ^{57}Fe Mössbauer spectroscopy provided us with detailed information on the local vibrational, electronic and magnetic state of iron in the studied compounds, which enabled us to draw novel conclusions concerning the magnetic, electronic or defect structure of these systems. The further detailed exploration of the temperature dependence of the local state of iron in these compounds is promising because it may reveal correlations between the macroscopic MR properties and the local electronic and/or magnetic state of iron, which hopefully will bring us closer to the understanding of the possible microscopic backgrounds of the CMR effect in general.

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